Effects of Sodium Lauryl Sulfate on Coelectrodeposition of Nickel and Fly Ash Particles

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Abstract- The nickel/fly ash composite plating was realized in the presence of sodium lauryl sulfate (SLS) as a stimulator in Watts bath. Morphology of coating was studied by SEM; microhardness by Hardness tester and cathodic behaviors of additives and corrosion resistance by electrochemical polarization. The results showed that sodium lauryl sulfate facilitated the deposition of fly ash onto the nickel matrix. At fly ash 50 g/L, the amount of fly ash in the deposit was 16.5% at lauryl sulfate 10 g/L, compared with 8.2% at lauryl sulfate 0.1 g/L. The micro-hardness and corrosion resistance of coating increased with increasing fly ash powder in the deposit as well as lauryl sulfate concentration in the bath. The microhardness of the coating was HV 210 at fly ash 5 g/L without lauryl sulfate; in the presence of 100 g/L of fly ash and 10 g/L of lauryl sulfate, this value increased to HV 490 (*cf.* HV 198 for pure nickel coating).

Keywords - Composite plating, fly ash, micro-hardness, corrosion resistance, stimulator

I. INTRODUCTION

Composite plating has been used to improve surface characteristics of materials such as mechanical and electrochemical properties[1,2]. Composite coating properties are dependent on characteristics and amount of inert particles in the deposit [3-5]. The amount of particles in the deposit can be increased by their concentration in the electrolyte, also by controlling the plating conditions (e.g. current density, stirring speed, particles concentration). Some researchers have attempted to increase particle amount in deposit by using additives so called stimulators (promoters). Organic compounds used as a stimulator were described in the work of Shrestha [6]. In their study, an amount of SiC in Ni/SiC composite plating reached 45% (volume fraction) using AZTAB (Azobenzene moiety) at a concentration of 0.6 g/L. Inorganic compounds as a Co-cation or rare earth element salts were also studied as a stimulator in composite plating [5,7-10].

The surfactants as stimulators can vary compositions, microstructure of matrix as well increase the inert particles amount in deposit [11], because surfactants are able to improve the stability of a suspension by increasing the wettability and surface charge of suspended particles, and also enhance the electrostatic adsorption of suspended particles on the cathode surface by increasing their positive charge [12-15].

Sodium lauryl sulfate (SLS) with the formula $CH_3(CH_2)_{11}OSO_3Na$ is an organ-sulfate consisting of a 12-carbon tail attached to the sulfate group giving the material the amphiphilic properties. As SLS is derived from inexpensive coconut and palm oils, it is a common component of many domestic cleaning products. Also, SLS is an anionic surfactant used as a wetting reagent or a brightener for Ni plating. On the other hand, SLS was studied as a stimulator for Ni composite plating [10,15-21].

Fly ash (FA) – a byproduct of thermal electric power plants has been used for manufacturing of alloys composites [22], polymer composites [23-24], zinc electroplating composite coatings [25].

II. EXPERIMENTAL

A. Plating bath

Electrodeposition was realized in Watts bath, its compositions (g/L) were: $Ni_2SO_4.6H_2O$: 250; $NiCl_2.6H_2O$: 40; H_3BO_3 : 40; inert particles FA (6-8 μ m): 5-100; surfactant SLS: 0.1-10.

B. Materials and study methods

Plating materials, chemicals, samples preparation before and after plating; also the determination of coating thickness by Axiover 40 MAT (Carl Zeiss); the hardness by IndentaMet 1106 (Buehler); corrosion resistance of coating by electrochemical polarization with PARSTAT 2273 and FA amount in deposit by SEM Jeol 6490-2300 in combination with EDX and/or gravimetric method were described in our previous paper [26,27].

C. Plating conditions

From our research²⁷, the maximum amount of FA included in the composite deposit was achieved at a current density of 2 A/dm^2 , in a plating solution of pH4, at a temperature of 50°C and a stirring speed of 250 rpm. These conditions were repeated again in present work.

III. EXPERIMENT AND RESULT

A. Cathodic polarization study

The cathodic behavior of different constituents of plating bath was studied by cathodic polarization (Fig. 1) at a scan rate of 0.1 mV/s. It can be seen that the cathodic polarization curves of pure nickel (curve 1) and nickel composite plating with different FA angd SLS concentrations in the solution (curve 2-5) were nearly coincided. As the cathodic potentials moved towards negative, the deposition rate of Ni-cation increased to - 0.4 V. Then the deposition of Ni reached a stable state at a current density of 10^{-4} A/cm² in a potential range from - 0.4 to - 0.7 V; below potential of - 0.7 V, hydrogen gas appeared.



Figure 1. Cathodic polarization curves of nickel deposition at scan rate: 0.1 mV/s with different additives concentrations in the plating bath (g/L): (a) SLS: 0; FA: 20, 50, 100; (b) SLS: 10; FA: 5, 20, 50, 100

The similarity of cathodic polarization curves revealed that FA as well as SLS had no effect on the deposition of Ni cation, however, at the FA concentration of 100 g L^{-1} , the deposition of Ni cation exhibited a slowdown at a low range of the deposit current density (curve 4, Fig. 1a). This may be due to blocking effect of the FA particles at the cathode surface.

As a wetting reagent for Ni plating, SLS was capable to adsorb on the cathode surface, this adsorbed layer decreased the deposit rate of Ni cations and reinforcing particles. This affected the crystallization kinetics of Ni and Ni/FA composite plating resulting in the different coating microstructures (Fig. 2b,c).

B. Surface morphology study

The morphology of Ni and Ni-composite plating was studied by SEM/EDX and shown in Fig. 2. It has been observed that the microstructure of Ni coating became finer even in the presence of only FA as showed in Fig 2b. In the simultaneous presence of FA (50 g/L) and SLS (10 g/L), the microstructure of coating was changed (Fig. 2c). In this case, the size of coating crystalline was in the range of 2-3 μ m similar to the one in the case without SLS (Fig. 2b). However, it was observed that in the presence of SLS the microstructure of coating seemed being denser. In some areas the crystals aggregated forming granular forms (Fig. 2c).



Figure 2. SEM microphotographs of Ni and Ni/Fa composite at different additives concentrations (g/L): (a) Ni; (b) Ni/FA50; (c) Ni/FA50-SLS10; (bar = 10 μm)

C. Amount of FA in the deposit

The separate and total amount of SiO_2 and Al_2O_3 was determined and the results were illustrated in Fig. 3. It can be seen that the FA amount in the deposit increased with increasing FA and SLS concentrations in plating bath. FA amount of 18.8% (mass fraction) in the deposit was obtained at FA 100 g/L and SLS 10 g/L. Experiments with higher FA and SLS concentrations were not carried out because at these concentrations the plating bath became viscous and turbid.



Figure 3. Relationship between individual and total amount of SiO₂, Al₂O₃ (mass fraction, %) in deposit with different compositions of plating bath (FA: 5-100 g/L; SLS: 0.1-10 g/L)

Measurements of the surface charge of FA particles in the solution with different pH values were performed with Zeta Phoremeter CAD - Z3110 (French). The results indicated a low positive charge (zeta potential of ~ 10 mV) at the FA surface in the solution with pH < 4.5 corresponding to pH of Ni plating solution. It agreed with data from reference [8].

Owing to the positive charge, according to the Guglielmi' model [28], the FA particles were adsorbed on the cathode, thus embedded in growing metal deposit. However, due to the low positive charge as well as adsorption competition with the Ni-cations, the amount of FA deposited on the cathode was not high.

As shown by Lee [29], the magnitude of the zeta potential of the particles was a measure of particles interaction. Since the zeta potential of the FA was less than 10 mV – close to zero, the FA suspension was unstable and particle aggregation occurred resulting low amount of reinforcing particles in deposit [12,13]. However, the surfactant is able to improve the dispersion of inert particles facilitating the particles moved and bound to cathode; hence their amount in deposit was increased.

In the presence of a surfactant, the reinforcing particles are also affected by interacting with the surfactant [30]. It was plausible that a suitable surfactant could improve the stability of particle suspension and the charge of suspended particles and thus enhancing the electrostatic adsorption of the suspended particles on a cathode.

The adsorption of the SLS anions on the surface of positively charged FA particles formed a negative charge cloud on which Ni-cations could be adsorbed. This adsorption increased positive charge of FA particles accelerating both the electrophoretic movement of the FA particles towards the cathode and electrostatic attraction between the FA and the cathode. This dual action facilitated the codeposition of FA onto the cathodic surface. Since the presence of Ni cations on the surface of FA particles contributed to a high amount of FA codeposited on the cathode, the reduction of metal ions was considered as the key step in the process of co- deposition [30,31].

Experimental results showed that in the presence of surfactant SLS, at the concentration of FA of 50 g/L, the amount of FA deposited in the coating increased 2.3 times (from 8% to 18.8% (mass fraction); this amount increased by increasing simultaneously both FA and SLS concentration.

D. Micro-hardness of coating

Micro-hardness of the Ni/FA composite plating was summarized in Fig. 4. The values in the figure were an average of 10 different measured points for one sample. Clearly that micro-hardness of the Ni/FA composite coating was higher than that of pure Ni. As the FA concentration in the solution increased, the hardness of composite coating increased due to a higher amount of FA particles co-deposited in the coating. Furthermore, the amount of FA powder in coating increased with increasing SLS concentration resulting in an increase of the micro-hardness (Fig. 4). For example, comparison of micro-hardness of the coatings obtained at a given FA powder concentration (50 g/L) but different SLS concentrations (0 and 10 g/L) indicates a good increase in micro-hardness. Indeed, it has been found that the micro-hardness was HV 270 for 0 g/L of SLS, and HV 460 at 10 g/L for SLS or 70.4% increase (Fig.5). Also, at the two extremes in Fig. 5 if the micro-hardness of Ni coating (HV 198) compared with that of composite coating achieved at FA concentration of 100 g/L and SLS of 10 g/L (HV 490), the increase was 147% (or 2.5 times higher.



Figure 4. The Microhardness of Ni/FA composite coating (plating bath compositions: FA: 5-100 g/L; SLS: 0.1-10 g/L).

A close examination of the microstructure of composite coating (Fig. 2b, 2c) revealed that some granular structures formed in the presence of SLS but the Ni maintained at a similar size. Although SLS affects the microstructure, however the enhancement of the micro-hardness was mainly dependent on the FA content in the coating.

E. Electrochemical property of coating

Fig. 5 shows typical anodic polarization curves of the Ni and Ni/FA composite coating in the absence and presence of SLS in 3.5% NaCl solution. From these curves, the electrochemical parameters of the Ni and the composite coatings were calculated and shown in the Table 1.



Figure 5. Anodic polarization curves of composite plating in 3.5% NaCl (scan rate: 0.1 mV/s)

Sample	$E_{\rm corr}/({\rm mV})$	$J_{\rm corr}/(\mu {\rm A/cm}^2)$	$E_{\rm pit}/({\rm mV})$	$J_{\rm pit}/({\rm mA/cm^2})$	$R_{\rm p}/({\rm k}\Omega{\rm cm}^2)$
Ni	-280	1.048	105	0.775	18.85
Ni/FA50	-228	0.576	92	0.361	38.64
Ni/FA50/SLS0.1	-226	0.520	102	0.287	44.69
Ni/FA50/SLS0.2	-204	0.421	107	0.277	61.86
Ni/FA50/SLS0.5	-202	0.381	112	0.234	59.64
Ni/FA50/SLS5.0	-194	0.272	71	0.176	81.56
Ni/FA50/SLS10.0	-160	0.181	79	0.145	112.86
Ni/FA100/SLS5.0	-132	0.138	89	0.025	91.39

Note: Each parameter is average values measured at least on the three samples

 E_{corr} : Corrosion potential, J_{corr} : Corrosion current density, E_{pit} : Pitting potential, J_{pit} : pitting current density

From data listed in the Table 1, clearly that composite plating, particularly in the presence of SLS surfactant, was nobler in comparison with Ni-coating. In the NaCl solution, corrosion current density J_{corr} ($\mu A/cm^2$) of Ni was 1.048; of Ni-FA50: 0.576; of Ni-FA50+SLS10: 0.181; polarization resistance R_p ($k\Omega cm^2$) was 18.85 for Ni; 38.64 for Ni-FA50 and 112.86 for Ni-FA50+SLS10, respectively. The corrosion resistance of Ni-FA composite is 2 times higher than that of Ni, whilst Ni-FA+SLS nearly 6 time higher. It is thus proposed that the presence of reinforcing particles embedded in the Ni matrix resulted the finer microstructures and the narrower granular boundaries, even crevices, gaps, cracks and micro pores were filled by additives. These particulates act as an inert physical barrier against initiation and development of corrosion, hence improving the corrosion resistance [19].

By increasing FA concentration at a constant concentration of SLS, the corrosion resistance of composite increased, i.e. the corrosion current (J_{corr}) decreased and the polarization resistance (R_p) increased. For example, at a SLS concentration of 5 g/L, corrosion current density is 0.272 μ A/cm², polarization resistance is 81.56 k Ω cm² at FA powder of 50 g/L, and 0.138 μ A/cm² and 91.39 k Ω cm² at FA powder of 100 g/L.

In the case of a constant FA concentration, as the SLS concentration increased, the corrosion resistance of composite increased. For example, at 50 g/L of FA, 0.1 and g/L of SLS, the corrosion current density (J_{corr}) decreased from 0.520 to 0.181 μ A/cm² (~ 3 times). In all cases, the pitting potentials (E_{pit} /mV) and the pitting current densities (J_{pit} /mA/cm²) decreased. This again indicated enhancement of corrosion resistance of the composite coating.

IV.CONCLUSION

- As an anion surfactant, SLS did not affect on the cathodic kinetics of Ni cation. SLS exhibited a good stimulating effect to facilitate the co-deposition of FA in the Ni plating. The amount of FA in the deposit increased with the SLS concentrations. For example, at the FA concentration of 50 g/L, when SLS concentration of 0.1, 5 and 10 g/L, the FA powder amount in deposit was 8.2, 13.3 and 16.3 % (mass fraction), respectively.
- The microhardness and corrosion resistance of the Ni/FA composite coatings were improved by an increasing amount of inert particles in the deposit that in turn was dependent on the FA and SLS concentrations in solution. Under selected plating conditions (2 A/dm², 250 r/min, 50 °C), at 50 g/L of FA, 10 g/L of SLS, microhardness of the coating increased 2.4 times, corrosion resistance about 6 times in comparison with pure Ni coating.
- As a stimulator, SLS facilitated more FA deposited into Ni matrix, thus higher microhardness and better corrosion resistance of the coatings were achieved.

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